

# A Combined Theoretical and Experimental Study of Oxygen Vacancies in $\text{Co}_3\text{O}_4$ for Liquid-Phase Oxidation Catalysis

Amir Omranpour,\* Lea Kämmerer, Catalina Leiva-Leroy, Anna Rabe, Takuma Sato, Soma Salamon, Joachim Landers, Benedikt Eggert, Eugen Weschke, Jean Pascal Fandré, Ashwani Kumar, Harun Tüysüz, Martin Muhler, Heiko Wende, and Jörg Behler



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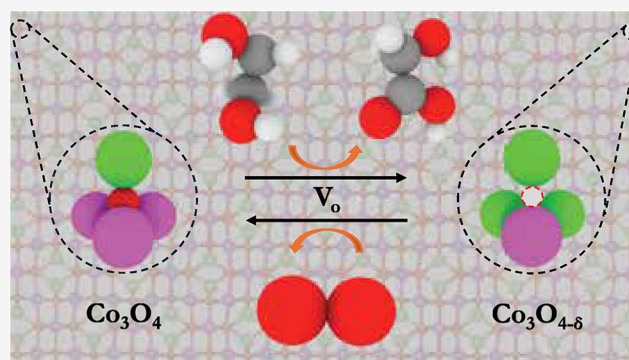


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**ABSTRACT:** In the present work, we investigate oxygen vacancies ( $V_{\text{O}}$ ) in  $\text{Co}_3\text{O}_4$ , both in the bulk phase and under liquid-phase ethylene glycol (EG) oxidation by combining theoretical and experimental techniques. Density functional theory (DFT) calculations for bulk  $\text{Co}_3\text{O}_4$  show that introducing an oxygen vacancy reduces two adjacent  $\text{Co}^{3+}$  ions to  $\text{Co}^{2+}$  and narrows the band gap. The newly formed  $\text{Co}^{2+}$  ions adopt high-spin configurations in distorted octahedral sites and remain stable in this state in *ab initio* molecular dynamics (AIMD) simulations at 300 K. Computed O and Co K-edge X-ray absorption spectra (XAS) for ideal and vacancy-containing  $\text{Co}_3\text{O}_4$  show excellent agreement with the experimental data and serve as references to analyze the liquid-phase ethylene glycol oxidation. The comparison with experimental K-edge spectra of fresh and postreaction catalysts shows that fresh samples resemble the vacancy-containing reference, whereas postreaction spectra shift toward the ideal reference. These results suggest that under liquid-phase ethylene glycol oxidation conditions,  $\text{Co}_3\text{O}_4$  becomes more oxidized rather than reduced, by refilling preexisting oxygen vacancies. This is further supported by the observation that higher  $\text{O}_2$  pressures increase the conversion and that the catalyst remains stable and active over several cycles.



## I. INTRODUCTION

Cobalt oxide is a mixed-valence transition metal oxide that has gained considerable attention due to its distinctive chemical, physical, and electronic characteristics. Among its various forms, the cobalt spinel phase  $\text{Co}_3\text{O}_4$  has been extensively investigated owing to its outstanding performance in oxidation catalysis, particularly in selective hydrocarbon oxidation.<sup>1–3</sup> The coexistence of  $\text{Co}^{2+}$  and  $\text{Co}^{3+}$  ions in  $\text{Co}_3\text{O}_4$  gives rise to versatile redox activity, high catalytic performance, a tunable electronic structure, and intriguing magnetic behavior.<sup>4–12</sup> These properties make  $\text{Co}_3\text{O}_4$  an attractive material for a broad range of technological and industrial applications, including oxidation catalysis of alcohols,<sup>13</sup> water oxidation,<sup>14,15</sup> methane combustion,<sup>16</sup> and CO oxidation,<sup>17</sup> as well as in lithium-ion batteries and gas sensors.<sup>18</sup>

Structurally,  $\text{Co}_3\text{O}_4$  adopts a cubic  $Fd\bar{3}m$  spinel structure in which cobalt exists in two oxidation states,  $\text{Co}^{2+}$  and  $\text{Co}^{3+}$ . The  $\text{Co}^{2+}$  ions occupy the tetrahedral interstitial sites, while  $\text{Co}^{3+}$  ions reside in the octahedral sites within the face-centered cubic (FCC) lattice formed by oxygen anions (Figure 1). The crystal field splits the 5-fold degenerate d orbitals of cobalt into two distinct energy levels. As a result, high-spin  $\text{Co}^{2+}$  exhibits three unpaired d electrons, corresponding to an experimental

magnetic moment of  $3.26 \mu_{\text{B}}$ ,<sup>19</sup> whereas  $\text{Co}^{3+}$  adopts a low-spin configuration with a nearly quenched magnetic moment (Figure 1). At ambient temperature,  $\text{Co}_3\text{O}_4$  behaves as a paramagnetic semiconductor and becomes antiferromagnetic below 40 K,<sup>19</sup> mainly due to weak exchange interactions between neighboring  $\text{Co}^{2+}$  ions.  $\text{Co}_3\text{O}_4$  is an intrinsic p-type semiconductor; its charge transport between 220 and 400 K occurs predominantly via small-polaron hopping of holes, while in the 170–220 K range it proceeds through variable-range hopping.<sup>20,21</sup> The estimated experimental band gap of  $\text{Co}_3\text{O}_4$  is approximately 1.6 eV.<sup>22,23</sup>

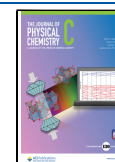
In heterogeneous catalysis,  $\text{Co}_3\text{O}_4$  is recognized for its effectiveness in the oxygen evolution reaction and, more recently, in alcohol oxidation. Increasing attention has been directed toward performing such oxidation reactions in the liquid phase,<sup>13,24</sup> often in alkaline solution, which enables

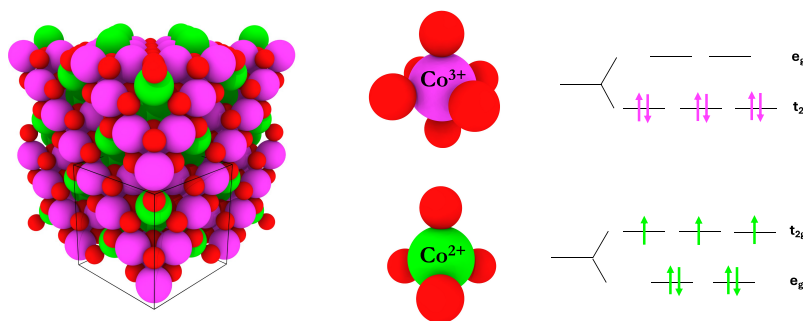
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**Figure 1.** Co<sub>3</sub>O<sub>4</sub> spinel (2 × 2 × 2) supercell. Co<sup>2+</sup> cations (green) occupy tetrahedral sites and Co<sup>3+</sup> cations (purple) occupy octahedral sites. The right side of the figure illustrates the corresponding crystal-field splittings: the lower-right diagram shows tetrahedrally coordinated Co<sup>2+</sup> in a d<sup>7</sup> configuration, and the upper-right diagram shows octahedrally coordinated Co<sup>3+</sup> in a d<sup>6</sup> configuration in the low-spin state.

milder reaction conditions and improved selectivity.<sup>25</sup> Nevertheless, the precise influence of the aqueous environment on Co<sub>3</sub>O<sub>4</sub> catalysis remains ambiguous, as both promoting and deactivating effects have been reported in computational<sup>26</sup> and experimental<sup>13,25</sup> studies. Another open question is the actual state of Co<sub>3</sub>O<sub>4</sub> under these conditions. Often, one starts from a “fresh” sample and assumes that the catalyst remains relatively stable during the catalytic process. However, questions such as how the local structure, oxidation state, and magnetic state of Co<sub>3</sub>O<sub>4</sub> change during the catalytic reaction have not yet been thoroughly answered.

On the other hand, oxygen vacancies (V<sub>O</sub>) in Co<sub>3</sub>O<sub>4</sub> are widely considered important for enhanced catalytic activity,<sup>27–32</sup> since they can increase the Co<sup>2+</sup>/Co<sup>3+</sup> ratio and create sites that are more easily oxidized or reduced during the reaction cycle. However, most of these conclusions are based on gas-phase reactions or static models that do not address the stability of vacancy-induced Co<sup>2+</sup> species under liquid-phase oxidation conditions.

For these reasons, in this work, we combined theoretical and experimental approaches to (i) identify the local electronic and magnetic response to a single V<sub>O</sub> in Co<sub>3</sub>O<sub>4</sub>, (ii) examine its stability at room temperature, and (iii) correlate these states with O K-edge measurements on fresh and postreaction samples tested in the liquid-phase ethylene glycol (EG) oxidation, to further clarify the actual role of oxygen vacancies in this particular catalytic process. It should be noted that this work does not focus on the mechanistic aspects of ethylene glycol oxidation in the liquid phase, and the interested reader is referred to the literature for detailed mechanistic studies.<sup>25,33,34</sup>

The structure of this work is as follows. First, we present density functional theory (DFT) results for ideal Co<sub>3</sub>O<sub>4</sub> and for Co<sub>3</sub>O<sub>4</sub> containing an oxygen vacancy, analyzing the changes in magnetic moments, density of states (DOS), and charge density differences (CDD). We then use *ab initio* molecular dynamics (AIMD) to test the finite-temperature stability of the vacancy-induced Co<sup>2+</sup> sites. Finally, we compare the calculated X-ray absorption spectra (XAS) (O K-edge and Co K-edge) with the experimental spectra of fresh and postreaction samples and discuss the implications for vacancy creation or healing under liquid-phase oxidation conditions.

## II. COMPUTATIONAL DETAILS

All electronic structure calculations, *ab initio* molecular dynamics simulations, and X-ray absorption spectroscopy calculations were carried out using the Vienna *Ab initio* Simulation Package (VASP)<sup>35,36</sup> (version 6.3.2) within the

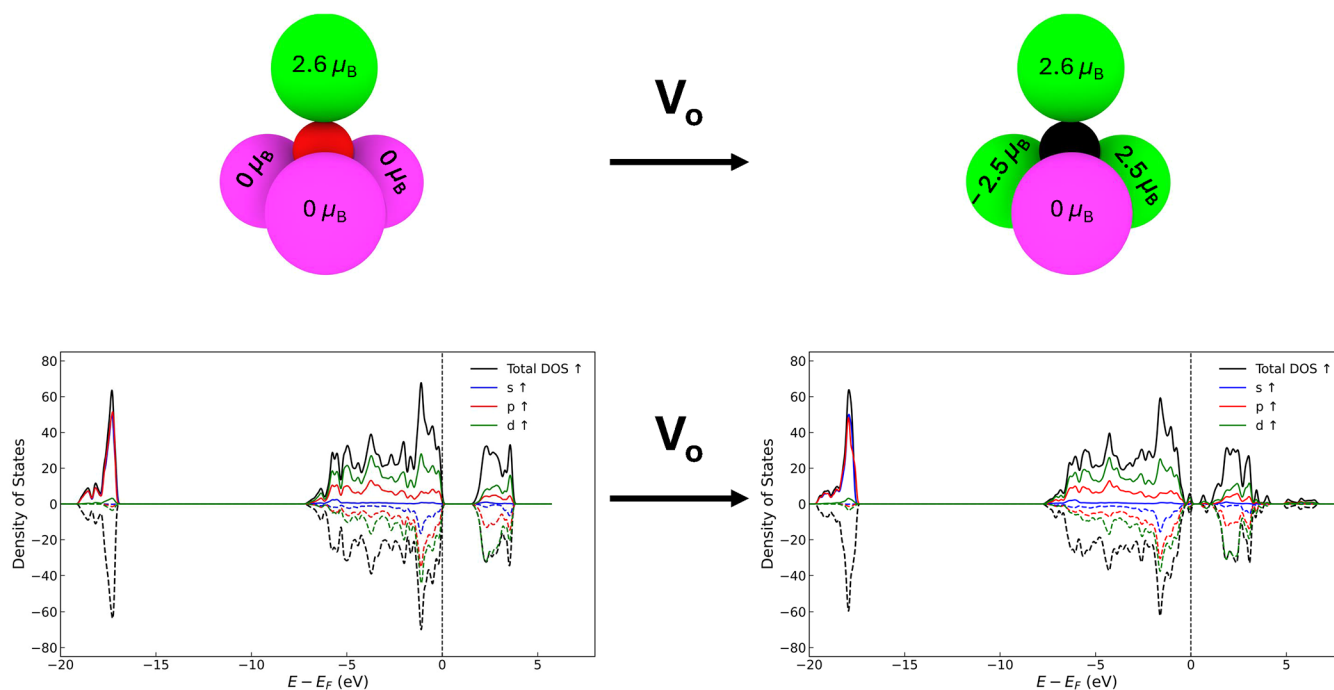
framework of spin-polarized density functional theory (DFT). Exchange–correlation effects and long-range dispersion interactions were described by the optPBE-vdW functional.<sup>37–39</sup> On-site Coulomb interactions among the Co 3d electrons were treated within the DFT + *U* formalism of Dudarev et al.,<sup>40</sup> employing an effective Hubbard parameter of  $U_{\text{eff}} = 2.43$  eV. We have shown previously that the aforementioned setting can reproduce the electronic and structural properties of Co<sub>3</sub>O<sub>4</sub> in good agreement with experimental and theoretical data.<sup>41</sup>

The interaction between valence electrons and ionic cores was represented by the Projector Augmented Wave (PAW) method,<sup>42</sup> as implemented by Kresse and Joubert.<sup>36,43</sup> Plane-wave expansions were performed with a kinetic energy cutoff of 500 eV. The Brillouin zone was sampled using a Monkhorst–Pack grid of 5 × 5 × 5 *k*-points for bulk Co<sub>3</sub>O<sub>4</sub>. The bulk Co<sub>3</sub>O<sub>4</sub> model contained 56 atoms in a cubic supercell of approximately 8 × 8 × 8 Å<sup>3</sup> (see refs 41,44). Gaussian smearing of 0.1 eV was applied for partial occupancies, and nonspherical contributions inside the PAW spheres were included. Electronic self-consistency was reached when the total energy difference between successive iterations was less than 10<sup>−6</sup> eV.

*Ab initio* molecular dynamics (AIMD) simulations of the bulk Co<sub>3</sub>O<sub>4</sub> were performed in the *NPT* ensemble using a Langevin thermostat.<sup>45</sup> The target temperature was held at  $T = 300$  K. A time step of  $\Delta t = 0.5$  fs was used for  $N_{\text{step}} = 100,000$  steps, yielding a total trajectory length of 50 ps. Forces and stresses were evaluated at each step with the same electronic settings as those above. The trajectories were generated at the  $\Gamma$  point.

The X-ray absorption spectra of the O K-edge and Co K-edge were computed on top of the converged spin-polarized DFT + *U* calculations described above. For both edges, we employed a core-excited final-state approach within VASP,<sup>46</sup> in which the absorbing atom is replaced by a PAW potential containing a 1s core hole, while all other atoms retain the ground-state potentials. The same optPBE-vdW functional,  $U_{\text{eff}} = 2.43$  eV, and plane-wave cutoff as in the ground-state calculations were used to ensure consistency. In order to minimize spurious interaction between periodic images of the core hole, the XAS was evaluated in the 56-atom Co<sub>3</sub>O<sub>4</sub> supercell, and the core hole was placed on the site of interest, i.e., lattice O or Co in octahedral/tetrahedral coordination.

Following recent theoretical XANES studies of Co<sub>3</sub>O<sub>4</sub> at the O K-edge<sup>11</sup> and at the Co K-edge,<sup>47</sup> the calculated spectra were postprocessed by convolution with an energy-dependent



**Figure 2.** Local structural and electronic response of  $\text{Co}_3\text{O}_4$  to the creation of an oxygen vacancy. Top: atomic configuration around the O site before (left) and after (right) removal of the central lattice O atom. In the ideal case, the O atom is coordinated to three low-spin  $\text{Co}^{3+}$  (purple,  $0 \mu_{\text{B}}$ ) and one high-spin  $\text{Co}^{2+}$  (green,  $2.6 \mu_{\text{B}}$ ). After the introduction of the O vacancy, two of the neighboring  $\text{Co}^{3+}$  ions are reduced to  $\text{Co}^{2+}$ , as indicated by their change to green and the appearance of finite magnetic moments, demonstrating local charge and spin redistribution. Bottom: spin-resolved, orbital-projected density of states (s, p, d) summed over all atoms. Solid lines denote the spin-up channel and dashed lines of the same color denote the spin-down channel (only spin-up is listed in the legend). In ideal  $\text{Co}_3\text{O}_4$  a clear band gap is present. After the vacancy is created, additional Co 3d character appears closer to the Fermi level, spin splitting becomes visible, and the band gap is narrowed.

Lorentzian to account for core-hole lifetime broadening and inelastic losses, and an additional Gaussian of constant width to mimic the experimental energy resolution. A rigid energy shift was then applied so that the intense pre-edge–edge feature of the calculated spectra coincides with the corresponding experimental peak. The spectra were baseline-corrected and normalized to unity in a common post-edge energy window, consistent with the experimental normalization to the edge jump. For more details on these postprocessing steps, the interested reader is referred to the above papers. For the Co K-edge, separate spectra were computed for tetrahedral  $\text{Co}^{2+}$  and octahedral  $\text{Co}^{3+}$  sites and combined in the stoichiometric ratio of 1:2 ( $\text{Co}_{\text{tet}}^{2+}/\text{Co}_{\text{oct}}^{3+}$ ), using the same broadening, alignment, and normalization protocol for all ideal and vacancy-containing configurations.

### III. EXPERIMENTAL DETAILS

#### III.I. Liquid-Phase Ethylene Glycol Oxidation

The experiments were performed in a batch reactor (Büchi) made of Hastelloy C-22, which is resistant to highly alkaline conditions. Catalytic reactions were conducted at  $120^\circ\text{C}$  under aerobic conditions (10 bar  $\text{O}_2$ ) with 0.65 M KOH and 0.025 or 0.325 M ethylene glycol (EG) for 6 h. Further details of the  $\text{Co}_3\text{O}_4$  synthesis procedure by coprecipitation, SBA-15 hard-templated (HT), and spray-flame can be found in earlier publications.<sup>25,48–50</sup> Standard characterization of the as-synthesized and postcatalysis of the hard-templated  $\text{Co}_3\text{O}_4$  samples, and of further catalytic activity determination can be found in a previous study.<sup>25</sup>

The products were analyzed by HPLC (Azura, Knauer) using an RID detector. The tested samples were obtained after 6 h and then treated with  $\text{H}_2\text{SO}_4$  to obtain pH 2 according to the necessary measurement conditions to perform analysis with an Aminex HPX-

87H Column (BioRad). Glycolic acid (GA), formic acid (FA), and oxalic acid (OA) were the only detected products. The carbon balance was  $100 \pm 2\%$ .

#### III.II. X-ray Absorption Spectroscopy

The O K-edge and Co  $L_{2,3}$ -edges X-ray absorption spectroscopy measurements were performed at the UE46-PGMI<sup>51</sup> beamline at the synchrotron BESSY II. The measurements at the O K-edge were carried out in the total electron yield (TEY) mode with linear horizontally polarized X-rays, while the Co  $L_{2,3}$ -edges (shown in Figure S2) were measured with circular polarized X-rays. The absorption coefficient was calculated using the signal of the photocurrent divided by the monitor signal. The sample background was corrected by subtraction, with a linear function applied in the pre-edge region of the data. Afterward, the data was normalized with respect to the edge jump. The experimental data shown in Figure 6 are the result of two averaged scans for each sample.

The Co K-edge XAS measurements<sup>52</sup> were performed at the SAMBA beamline of the SOLEIL synchrotron, operating with an electron beam current of 450 mA. The incident energy was selected by a Si (200) double crystal monochromator. The incident flux was ca.  $1 \times 10^{10}$  ph/s using a beam size of  $1 \text{ mm} \times 1 \text{ mm}$ . The Mn and Co K-edges were utilized to conduct measurements in the transmission mode. The samples were prepared by diluting the corresponding powder with cellulose in a pellet ( $\varnothing = 13 \text{ mm}$ ), followed by placement in the sample holder and sealing with Kapton tape. Calibration was performed using Mn and Co metal foils. The final spectra were normalized and processed using the Fastosh software (for more details, see ref 52).

#### III.III. Magnetometry

The magnetic properties were recorded with the vibrating sample magnetometer (VSM) option of a Quantum Design PPMS DynaCool. Temperature-dependent  $M(T)$  magnetization curves were recorded with the zero field cooled-field cooled (ZFC-FC) protocol between 5 and 300 K with an applied magnetic field of 0.1 T.

## IV. RESULTS AND DISCUSSION

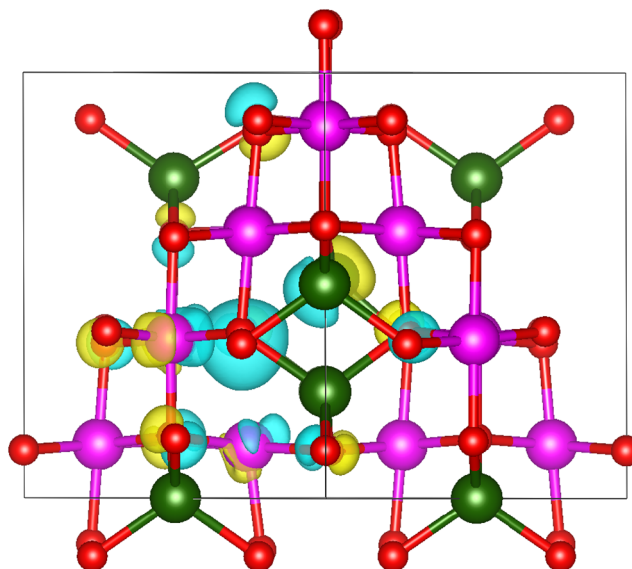
### IV.I. Oxygen Vacancies in Bulk $\text{Co}_3\text{O}_4$

**IV.I.I. DFT Calculations at 0 K.** First, DFT calculations for ideal  $\text{Co}_3\text{O}_4$  were carried out and used as a reference throughout this work. They reproduce the expected anti-ferromagnetic ground state and give a local magnetic moment of about  $2.6 \mu_{\text{B}}$  for the tetrahedral  $\text{Co}^{2+}$  sites and  $0 \mu_{\text{B}}$  for the octahedral  $\text{Co}^{3+}$  sites (see Table S1 in the Supporting Information). The optimized lattice constant obtained with the present DFT setup is  $8.156 \text{ \AA}$  and the band gap is  $1.61 \text{ eV}$ , both in good agreement with experimental reports<sup>19,23,53,54</sup> and consistent with previous DFT studies that are relevant to this work.<sup>11,47</sup> A more detailed validation of the accuracy of the current DFT settings is provided in ref 41.

Similar DFT calculations were performed for  $\text{Co}_3\text{O}_4$  containing a single oxygen vacancy. A comparison of the results with those of the ideal  $\text{Co}_3\text{O}_4$  case is shown in Figure 2. In the upper panels, the local atomic environment around the oxygen site is shown before and after removing the O atom. In the ideal structure (left), one O atom is coordinated to four Co atoms: three are  $\text{Co}^{3+}$  (purple) with  $0 \mu_{\text{B}}$  and one is  $\text{Co}^{2+}$  (green) with  $2.6 \mu_{\text{B}}$ . After the oxygen vacancy is introduced (right), two of the  $\text{Co}^{3+}$  cations adjacent to the vacancy are reduced to  $\text{Co}^{2+}$ , which are highlighted by the appearance of local magnetic moments (and are shown in green). This indicates that removing a single atom redistributes charge and spin in the neighborhood of the vacancy, converting previously nonmagnetic, low-spin  $\text{Co}^{3+}$  sites into high-spin  $\text{Co}^{2+}$  sites. In addition, the two newly formed high-spin  $\text{Co}^{2+}$  ions adopt opposite spin directions. Table S2 in the Supporting Information provides the detailed orbital-resolved spin moments (*s*, *p*, *d*) and the resulting total local magnetic moment for every ion in the  $\text{Co}_3\text{O}_4$  supercell. It may be worth adding that the existence of a finite X-ray magnetic circular dichroism (XMCD) signal, as shown in Figure S2, may also be attributed to the uncompensated moments that arise when the  $\text{Co}^{2+}$  ions adopt a high-spin configuration.

The lower part of Figure 2 shows the spin-resolved orbital-projected density of states (DOS) for the two cases. The DOS is decomposed into *s*, *p*, and *d* contributions. For each orbital, the spin-up component is shown as a solid line, while the corresponding spin-down component is shown as a dashed line (only the spin-up curves are listed in the legend, but each has a dashed counterpart in the plot). For the ideal  $\text{Co}_3\text{O}_4$  structure, the DOS displays a clear band gap and the *d* states around the Fermi level are nearly spin-compensated, which is consistent with low-spin  $\text{Co}^{3+}$ . After the oxygen vacancy is introduced, additional *d*-character appears closer to the Fermi level and the spin-up and spin-down *d* curves no longer perfectly overlap; this is indicative of the reduction of neighboring  $\text{Co}^{3+}$  to high-spin  $\text{Co}^{2+}$  and the associated local spin polarization. At the same time, the onset of unoccupied states shifts so that the band gap is narrowed compared to the ideal case.

Figure 3 shows the charge density difference (CDD)  $\Delta\rho = \rho_{V_{\text{O}}} - \rho_{\text{ideal}}$  for  $\text{Co}_3\text{O}_4$  after removal of one lattice oxygen atom. In this color scheme, cyan denotes electron density depletion and yellow denotes electron density accumulation. A large cyan lobe is located at the position of the missing O atom and stretches along the former Co–O bonds, confirming that negative charge is removed from that region when the oxygen is taken out. The yellow isosurfaces appear mainly on the neighboring Co cations and along some of the remaining Co–

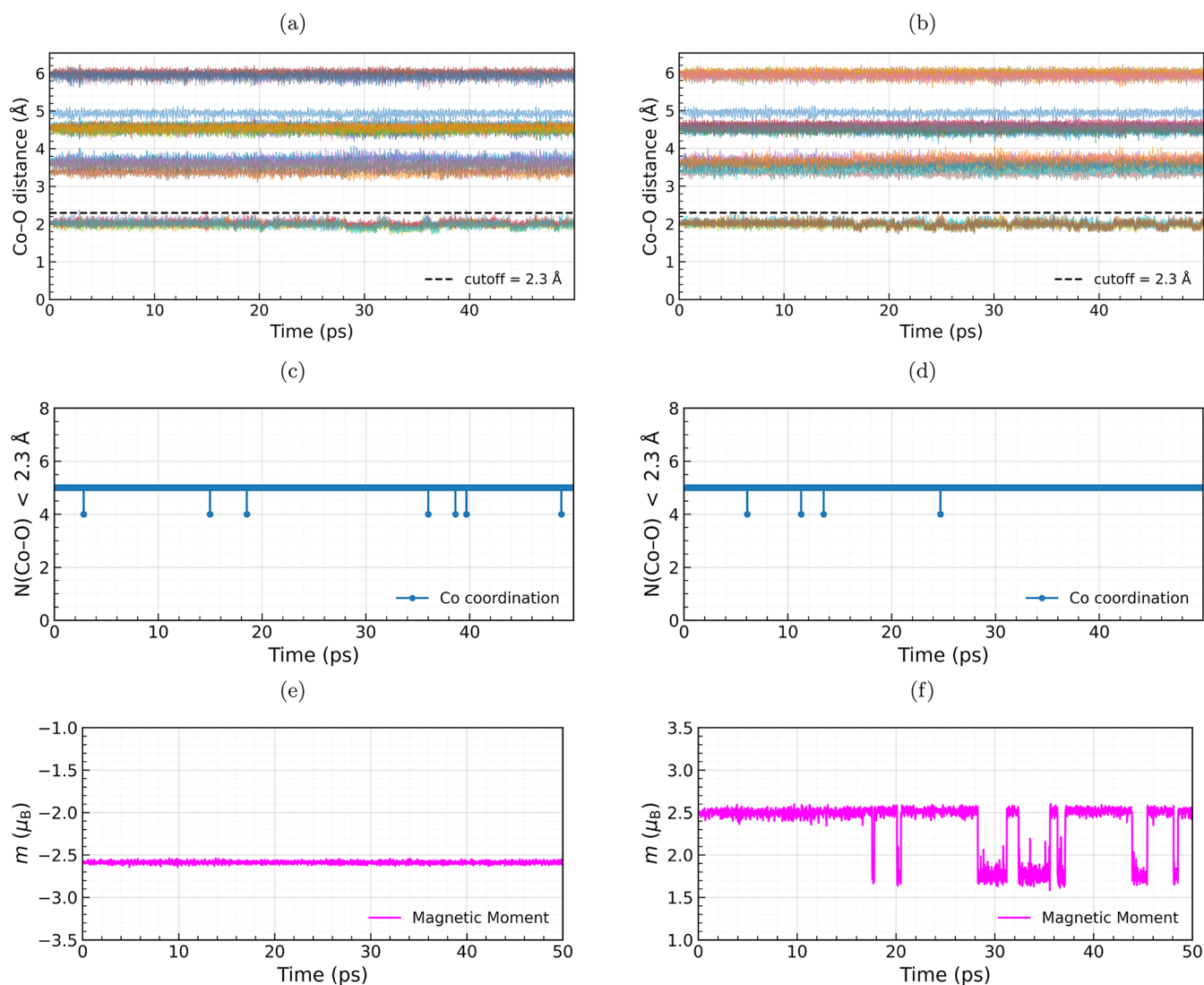


**Figure 3.** Charge density difference  $\Delta\rho = \rho_{V_{\text{O}}} - \rho_{\text{ideal}}$  for  $\text{Co}_3\text{O}_4$  with a single oxygen vacancy. Cyan isosurfaces indicate negative charge depletion at the vacancy site and along the former Co–O bonds, while yellow isosurfaces show negative charge accumulation on neighboring Co cations and nearby O atoms, showing redistribution of the electrons left behind by the removed oxygen.

O bonds, which shows where and how the electrons are redistributed. This indicates that due to the O vacancy, electron density is reallocated from the vacancy site itself and transferred to the adjacent Co–O units, thus locally reducing the nearby Co ions and polarizing the surrounding framework.

**IV.I.II. The Role of Finite Temperature.** An important question for their role in catalysis is whether the  $\text{Co}^{2+}$  species next to the oxygen vacancy in the optimized DFT structures remains stable at finite temperatures when the lattice is allowed to fluctuate. In the relaxed  $V_{\text{O}}$  structure the two Co atoms adjacent to the vacancy are 5-fold coordinated (distorted octahedral) and carry a high-spin moment of about  $2.5 \mu_{\text{B}}$ . To test the thermal stability of this configuration, we performed an *ab initio* molecular dynamics simulation at 300 K starting from the relaxed  $V_{\text{O}}$  structure and propagated the system for 50 ps with a 0.5 fs time step. In the ideal  $\text{Co}_3\text{O}_4$  unit cell, there are eight  $\text{Co}^{2+}$  sites, and here we label the two vacancy-induced  $\text{Co}^{2+}$  as Co 9 and Co 10. The analysis is shown in Figure 4, where the left column (panels (a), (c), (e)) corresponds to Co ion 9 and the right column (panels (b), (d), (f)) corresponds to Co ion 10.

Panels (a) and (b) report the time evolution of the distances between each of these Co atoms and all O atoms in the unit cell. In both cases five Co–O distances stay clustered in the  $1.9\text{--}2.2 \text{ \AA}$  range for the entire 50 ps trajectory, while all other O atoms remain clearly farther away ( $>3 \text{ \AA}$ ). This indicates that neither of the two Co atoms regains a full 6-fold octahedral environment, nor does it collapse to adopt a lower coordination. To quantify this, panels (c) and (d) show the corresponding coordination numbers as a function of time, obtained by counting all O atoms within  $2.3 \text{ \AA}$  of the Co ion. For both  $\text{Co}^{2+}$  sites, the coordination number is essentially constant at 5 throughout the trajectory, with only a few very short-lived drops to 4. These brief excursions coincide with a temporary elongation of one Co–O bond across the  $2.3 \text{ \AA}$  cutoff and do not persist. Importantly, we never observe an



**Figure 4.** Time evolution of the two Co ions next to the oxygen vacancy, followed over a 50 ps *ab initio* MD trajectory at 300 K. Panels (a, b) track all Co–O distances for Co 9 and Co 10. Each ion holds on to five short Co–O bonds in the range of 1.9–2.2 Å, while all remaining oxygen atoms are found beyond a distance of 3 Å. Neither site ever regains an octahedral environment; the 5-fold coordination imposed by the vacancy remains intact. The coordination number plots in panels (c, d), obtained by counting O atoms within 2.3 Å cutoff, show the same scenario: both Co<sup>2+</sup> centers fluctuate around a value of 5, with only momentary drops to 4 and never any increase to reach 6. This configuration is therefore structurally stable at 300 K. Panels (e, f) follow the magnetic moments. Co 9 stays essentially locked in its high-spin state at roughly 2.5 μ<sub>B</sub> for the entire simulation. Co 10 shows occasional, very short-lived excursions down to 1.6–2.0 μ<sub>B</sub>, but always returns to the same high-spin value and never displays anything resembling a spin collapse. Overall, these results show that the oxygen vacancy produces two Co<sup>2+</sup> ions that remain both structurally and magnetically robust over a full 50 ps trajectory at 300 K.

increase to 6, i.e., no spontaneous reoccupation of the vacant site or rearrangement into a regular octahedron occur at 300 K. We therefore conclude that the vacancy-induced, 5-fold coordinated Co<sup>2+</sup> environment is structurally stable on the 50 ps time scale at room temperature and that any further structural change would require overcoming a higher barrier than what thermal fluctuations at 300 K can provide.

Panels (e) and (f) track the local magnetic moments of the same two Co ions. Co 9 (panel (e)) retains a nearly constant moment of  $\sim -2.5 \mu_{\text{B}}$  over the whole simulation, which is exactly the value found in the static DFT calculation and is characteristic of high-spin Co<sup>2+</sup>. Co 10 (panel (f)) behaves similarly for most of the simulation but exhibits several short intervals in which its moment drops to about 1.6–2.0 μ<sub>B</sub>. These episodes are transient and the moment always recovers to  $\approx 2.5 \mu_{\text{B}}$ ; at no point does the spin collapse to  $\sim 0 \mu_{\text{B}}$ , i.e., we

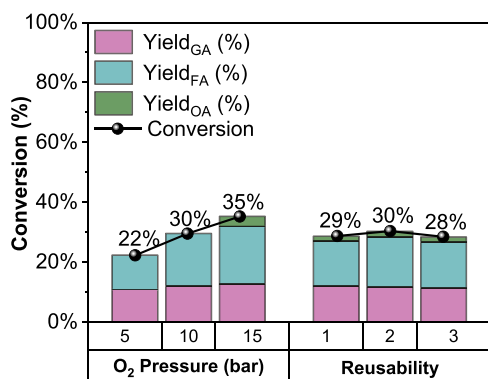
do not observe a conversion back to low-spin Co<sup>3+</sup>. Because the number of Co–O neighbors does not change in a permanent way, these short drops in the magnetic moment cannot be attributed to a change of site or oxidation state. The more plausible explanation is that the Co ion remains Co<sup>2+</sup> in the same 5-fold environment, but thermal motion at 300 K slightly perturbs the local bonds; consequently, the calculated moment briefly decreases before returning to its equilibrium value. Thus, these are small temperature-driven oscillations of the moment rather than a genuine Co<sup>2+</sup> → Co<sup>3+</sup> transition or a structural rearrangement. Overall, the AIMD confirms that the oxygen-vacancy-induced Co<sup>2+</sup> sites are both *structurally* and *magnetically* robust at 300 K on the 50 ps time scale. It may be worth noting that electron paramagnetic resonance (EPR) studies on Co<sub>3</sub>O<sub>4</sub> platelets used for the partial oxidation of cinnamyl alcohol have likewise reported a surface-near

distorted high-spin  $\text{Co}^{2+}$  center, tentatively attributed to a distorted octahedral or tetrahedral site.<sup>32</sup>

## IV.II. Liquid-Phase Ethylene Glycol Oxidation

### IV.II.I. Catalytic Activity, Selectivity, Reusability, and the Impact of $\text{O}_2$ Pressure.

Figure 5 summarizes the relation



**Figure 5.** Effect of the  $\text{O}_2$  pressure and catalyst reusability for the hard-templated  $\text{Co}_3\text{O}_4$  sample in the liquid-phase oxidation of ethylene glycol. The left part shows the EG conversion and the corresponding yields/selectivities of glycolic acid (GA), formic acid (FA), and oxalic acid (OA) after 6 h at 120 °C, 0.325 M EG, and 0.65 M KOH for  $\text{O}_2$  pressures of 5, 10, and 15 bar. The right part shows EG conversion and GA/FA/OA yields/selectivities for three consecutive runs at 10 bar  $\text{O}_2$  under the same conditions, which illustrates the stability and reusability of the hard-templated sample.

between the  $\text{O}_2$  pressure as well as catalyst reusability for the activity and selectivity of the hard-templated  $\text{Co}_3\text{O}_4$  samples during liquid-phase ethylene glycol oxidation. The left part of the figure shows the ethylene glycol conversion and the yields of glycolic acid (GA), formic acid (FA), and oxalic acid (OA) after 6 h at 120 °C, 0.325 M EG, and 0.65 M KOH for  $\text{O}_2$  pressures of 5, 10, and 15 bar. The total height of each bar corresponds to the overall EG conversion; the colored segments show individual yields of GA, FA, and OA, respectively. By increasing the  $\text{O}_2$  pressure, the ethylene glycol

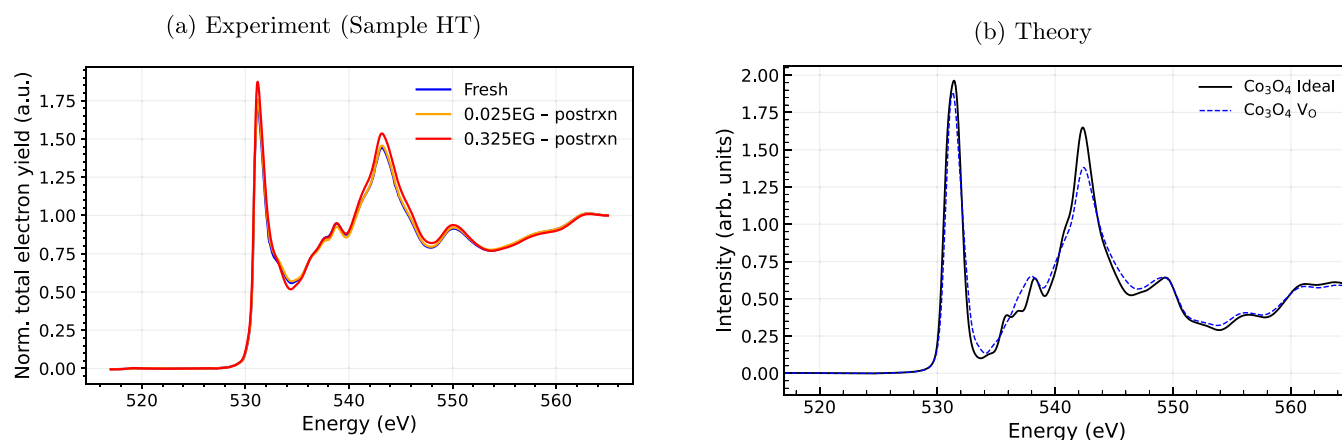
conversion and the absolute yields of the oxidation products increase. The selectivity pattern changes only moderately: the glycolic acid (GA) yield does not increase substantially with increasing  $\text{O}_2$  pressure, while the relative contributions of formic acid (FA) and oxalic acid (OA) are enhanced, i.e., at higher  $\text{O}_2$  pressure, more strongly oxidized products are favored.

The right part of Figure 5 shows the conversion and product yields for three consecutive reaction cycles, which are carried out at 10 bar  $\text{O}_2$  under the same alkaline conditions. Both the overall conversion and the relative distribution among GA, FA, and OA remain mainly unchanged in consecutive cycles. Thus, it can be suggested that the hard-templated  $\text{Co}_3\text{O}_4$  catalyst is not only active and selective but also structurally stable and reusable under the alkaline,  $\text{O}_2$ -rich liquid-phase reaction environment. Further details on the aforementioned experiments can be found in a recent publication.<sup>25</sup>

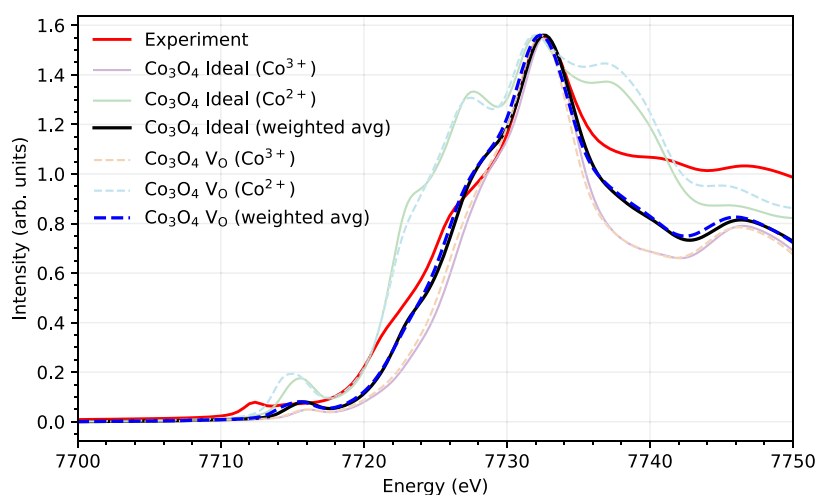
Figure S4 illustrates the influence of the ethylene glycol (EG) concentration on conversion and product yields in the liquid-phase oxidation of EG over hard-templated  $\text{Co}_3\text{O}_4$ . A lower EG concentration leads to higher conversion, as expected. However, the yield of glycolic acid does not show a significant increase; instead, glycolic acid is further oxidized to formic acid.

### IV.II.II. X-ray Absorption Spectroscopy. IV.II.III. O K-Edge X-ray Absorption Spectroscopy.

Figure 6 shows the evolution of the K-edge of  $\text{Co}_3\text{O}_4$  for the hard-templated sample (HT) and how this evolution compares with our theoretical reference spectra. Panel (a) corresponds to the HT sample and contains three curves: the fresh catalyst (blue), the 0.025 EG postreaction spectrum (orange), and the 0.325 EG postreaction spectrum (red). Panel (b) shows the theoretical O K-edge for the ideal  $\text{Co}_3\text{O}_4$  structure (black) and for  $\text{Co}_3\text{O}_4$  containing a single oxygen vacancy (blue). Although the discussion in this work focuses on the HT sample, the corresponding experimental spectra for the spray-flame synthesized (SF) and coprecipitated (CP) samples, which exhibit the same trends, are also provided in the Supporting Information (Figure S1).



**Figure 6.** O K-edge X-ray absorption spectra (XAS) of  $\text{Co}_3\text{O}_4$  for the hard-templated sample (HT) before and after liquid-phase ethylene glycol oxidation compared with theoretical reference spectra. Panel (a) shows the experimental spectra for the fresh catalyst (blue), the 0.025 EG postreaction state (orange), and the 0.325 EG postreaction state (red). Panel (b) displays the theoretical O K-edge spectra for the ideal  $\text{Co}_3\text{O}_4$  spinel (black) and for a structure containing a single oxygen vacancy (blue). The reduced intensity of the first two peaks in the vacancy-containing system closely resembles the experimental fresh-state spectrum, while the ideal calculated spectrum matches the oxidized 0.325 EG postreaction data. The same spectral trend is observed for the SF and CP samples (see Figure S1), all of which indicate that oxygen-vacancy-type defects are healed.



**Figure 7.** Co K-edge XANES of  $\text{Co}_3\text{O}_4$  compared to site-resolved theoretical spectra. The experimental spectrum of the fresh hard-templated sample<sup>52</sup> (red) serves as the reference. Calculated spectra for octahedral  $\text{Co}^{3+}$  and tetrahedral  $\text{Co}^{2+}$ , along with their stoichiometric weighted average, reproduce the experimental spectra in excellent agreement. For the  $\text{Co}_3\text{O}_4$  with  $\text{V}_\text{O}$ , the weighted spectrum exhibits a slight shift toward lower energy and a less steep edge, which suggests an increased  $\text{Co}^{2+}$  character. This modest but consistent change supports the interpretation of vacancy-induced local reduction, in agreement with the density of states (DOS), charge density difference (CDD), and O K-edge analyses as previously discussed.

For the HT sample, the trend is the same as for the other two catalysts. The fresh material has a slightly weaker intensity at the onset of the absorption, that is, in the first and second features of the O K-edge at 531 and 543 eV, respectively. After reaction, the spectra sharpen and the intensities at both the 531 eV peak and the 543 eV rise. This effect is most pronounced for the 0.325 EG postreaction state. The SF and CP samples follow the same behavior, as shown in Figure S1, therefore all three synthesis routes lead to catalysts that move in the same spectral direction when they are exposed to the liquid-phase ethylene glycol oxidation conditions.

The comparison with the calculated spectra in panel (b) makes this interpretation quantitative. In the calculation, the ideal  $\text{Co}_3\text{O}_4$  has two relatively strong and well separated low energy features, because the O 2p states are strongly hybridized with Co 3d states in a fully oxidized spinel environment. When an oxygen vacancy is introduced, the theoretical spectrum loses intensity in exactly those first two features.

The experimental fresh spectrum of HT in panel (a), together with the fresh spectra of SF and CP in Figure S1, resemble this vacancy-like calculated curve, while the 0.325 EG postreaction spectra resemble much more the ideal calculated curve with stronger first and second peaks at 531 and 543 eV, respectively. In other words, the catalytically treated samples evolve from a vacancy-perturbed O K-edge toward an almost ideal spinel-like O K-edge. Therefore, one may deduce that this correspondence between experiment and theory supports the idea that the liquid-phase reaction heals oxygen-vacancy-type defects and restores a more oxidized Co–O electronic structure. In other words, it can be suggested that there is a redox competition: ethylene glycol reduces the surface, while  $\text{O}_2$  in the reaction atmosphere reoxidizes it. This is further supported by the observation that higher  $\text{O}_2$  pressure leads to higher conversion and that the catalyst remains relatively stable and reusable over multiple reaction cycles (see Section IV.III).

As can be noted, the relative changes in the O K-edge XAS among the samples obtained by the three synthesis methods are different. Whether such differences can be associated with

different factors, e.g., different starting states (initial defectiveness or local Co–O environments) imposed by the synthesis methods, requires further investigation.

**IV.II.III. Co K-Edge X-ray Absorption Spectroscopy.** Figure 7 shows the Co K-edge XANES spectrum of  $\text{Co}_3\text{O}_4$  together with the site-resolved calculated spectra and their weighted sums. The experimental spectrum of the fresh hard-templated sample<sup>52</sup> is shown in red and serves as the reference (Co K-edge XAS for the postreaction samples are not yet available at the time of this publication). The calculated spectra for octahedral  $\text{Co}^{3+}$  and tetrahedral  $\text{Co}^{2+}$  in the ideal spinel, together with their stoichiometric weighted average, reproduce very well the edge position and the shape of the main rising feature in the experiment. As expected, the  $\text{Co}^{2+}$  curve is slightly lower in energy, the  $\text{Co}^{3+}$  curve is slightly higher and steeper, and the “ideal” (weighted averaged) spectrum lies between them, approaching the experimental curve.

The same site-by-site construction was then carried out for the structure containing an oxygen vacancy. In this case, the “ $\text{V}_\text{O}$ ” (weighted avg) spectrum is slightly shifted toward lower energy, and the edge is a bit less steep. Both effects are what one would expect if the vacancy locally increases the  $\text{Co}^{2+}$  character. The change, however, is modest: the  $\text{V}_\text{O}$  and ideal weighted spectra remain close, but the direction of the change is consistent with the density of states (DOS), charge density difference (CDD), and O K-edge results, which all indicate a vacancy-induced local reduction. In other words, the way the  $\text{V}_\text{O}$  spectrum shifts to slightly lower energy while getting broader, is fully consistent with the previously suggested picture that an oxygen vacancy leads to more  $\text{Co}^{2+}$  character.

**IV.II.III. Magnetometry.** In addition, magnetometry measurements have been performed on the hard-templated sample for the fresh and postcatalysis samples as a function of temperature (see Figure S3). Using the Curie–Weiss law, the temperature-dependent magnetization allows the calculation of the Curie constant and therefore the effective magnetic moment of the sample  $\mu_{\text{eff}}$  as described in Mugaraneza and Hallas.<sup>55</sup> Between the fresh sample and the two postcatalysis samples an average reduction of 5% of  $\mu_{\text{eff}}$  was observed.

Considering that in  $\text{Co}_3\text{O}_4$  only  $\text{Co}^{2+}$  contributes to the effective magnetic moment, the magnetometry results indicate that the  $\text{Co}^{3+}$  amount is reduced after catalysis, which is in agreement with the statements in the previous section regarding oxygen refilling and the oxidation of  $\text{Co}^{2+}$  to  $\text{Co}^{3+}$ .

## V. CONCLUSIONS

In the present work, we investigated oxygen vacancies in  $\text{Co}_3\text{O}_4$  both in the bulk phase and under liquid-phase ethylene glycol oxidation by combining spin-polarized DFT +  $U$ , *ab initio* molecular dynamics, and theoretical and experimental X-ray absorption spectroscopy (O and Co K-edge). Our main findings can be summarized as follows:

1. Introducing a single  $\text{V}_\text{O}$  into ideal bulk  $\text{Co}_3\text{O}_4$  reduces two neighboring octahedral  $\text{Co}^{3+}$  ions to high-spin  $\text{Co}^{2+}$  and narrows the band gap.
2. These newly formed  $\text{Co}^{2+}$  ions adopt similar local magnetic moments ( $\sim 2.5 \mu_\text{B}$ ) but in opposite directions.
3. They do not move into regular tetrahedral positions but remain in vacancy-perturbed, distorted octahedral environments with 5-fold coordination.
4. *Ab initio* MD at 300 K (50 ps) shows that both vacancy-induced  $\text{Co}^{2+}$  sites remain stable, structurally (the Co–O coordination stays at 5 with only brief fluctuations) and magnetically (the local moments stay in the  $\sim 2.5 \mu_\text{B}$   $\text{Co}^{2+}$  range). This suggests that any further structural transformation would require overcoming a higher activation barrier than what is accessible at 300 K in our simulations.
5. Comparison of the computed O K-edge spectra for ideal  $\text{Co}_3\text{O}_4$  and for  $\text{Co}_3\text{O}_4$  with  $\text{V}_\text{O}$  to the experimental O K-edge spectra of the as-prepared and postreaction catalysts shows that the fresh samples resemble the vacancy-containing calculation, while the spectra after liquid-phase ethylene glycol oxidation shift toward the ideal computed spectrum. This suggests that the reaction conditions tend to refill or heal oxygen-vacancy-type defects and drive the catalyst back toward a more oxidized, spinel-like state.
6. Consistently, magnetometry shows an average reduction of 5% in the effective magnetic moment after catalysis, suggesting a lower  $\text{Co}^{2+}$  content.
7. This picture is further supported by the fact that higher  $\text{O}_2$  pressures enhance EG conversion and that the catalyst remains stable and active across multiple reaction cycles.

## ■ ASSOCIATED CONTENT

### Data Availability Statement

The data generated and analyzed during the current study are available from the corresponding authors upon request.

### SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcc.5c08004>.

Additional O K-edge XAS spectra for SF and CP samples; Co  $L_{2,3}$ -edge XANES and XMCD for the HT sample; temperature-dependent magnetometry and Curie–Weiss analysis; and site-resolved DFT +  $U$  magnetic moments for ideal  $\text{Co}_3\text{O}_4$ , with an oxygen vacancy, and with SOC (Tables S1–S3) (PDF)

## ■ AUTHOR INFORMATION

### Corresponding Author

**Amir Omranpour** – *Lehrstuhl für Theoretische Chemie II, Ruhr-Universität Bochum, 44780 Bochum, Germany; Research Center Chemical Sciences and Sustainability, Research Alliance Ruhr, 44780 Bochum, Germany; [orcid.org/0009-0008-1500-3067](https://orcid.org/0009-0008-1500-3067); Email: [amir.omranpour@rub.de](mailto:amir.omranpour@rub.de)*

### Authors

**Lea Kämmerer** – *Faculty of Physics and Center for Nanointegration Duisburg-Essen (CENIDE), University of Duisburg-Essen, 47057 Duisburg, Germany; [orcid.org/0000-0002-3944-2696](https://orcid.org/0000-0002-3944-2696)*

**Catalina Leiva-Leroy** – *Laboratory of Industrial Chemistry, Ruhr-Universität Bochum, 44780 Bochum, Germany*

**Anna Rabe** – *Faculty of Physics and Center for Nanointegration Duisburg-Essen (CENIDE), University of Duisburg-Essen, 47057 Duisburg, Germany*

**Takuma Sato** – *Max Planck Institute for Chemical Energy Conversion, 45470 Mülheim an der Ruhr, Germany*

**Soma Salamon** – *Faculty of Physics and Center for Nanointegration Duisburg-Essen (CENIDE), University of Duisburg-Essen, 47057 Duisburg, Germany; [orcid.org/0000-0002-8661-6038](https://orcid.org/0000-0002-8661-6038)*

**Joachim Landers** – *Faculty of Physics and Center for Nanointegration Duisburg-Essen (CENIDE), University of Duisburg-Essen, 47057 Duisburg, Germany*

**Benedikt Eggert** – *Faculty of Physics and Center for Nanointegration Duisburg-Essen (CENIDE), University of Duisburg-Essen, 47057 Duisburg, Germany; [orcid.org/0000-0001-7739-3541](https://orcid.org/0000-0001-7739-3541)*

**Eugen Weschke** – *Helmholtz-Zentrum Berlin für Materialien und Energie (HZB), 12489 Berlin, Germany*

**Jean Pascal Fandré** – *Heterogeneous Catalysis and Sustainable Energy, Max-Planck-Institut für Kohlenforschung, 45470 Mülheim an der Ruhr, Germany; [orcid.org/0000-0003-2555-2429](https://orcid.org/0000-0003-2555-2429)*

**Ashwani Kumar** – *Heterogeneous Catalysis and Sustainable Energy, Max-Planck-Institut für Kohlenforschung, 45470 Mülheim an der Ruhr, Germany*

**Harun Tüysüz** – *Heterogeneous Catalysis and Sustainable Energy, Max-Planck-Institut für Kohlenforschung, 45470 Mülheim an der Ruhr, Germany; Catalysis and Energy Materials Group, IMDEA Materials Institute, 28906 Madrid, Spain; [orcid.org/0000-0001-8552-7028](https://orcid.org/0000-0001-8552-7028)*

**Martin Muhler** – *Laboratory of Industrial Chemistry, Ruhr-Universität Bochum, 44780 Bochum, Germany; [orcid.org/0000-0001-5343-6922](https://orcid.org/0000-0001-5343-6922)*

**Heiko Wende** – *Faculty of Physics and Center for Nanointegration Duisburg-Essen (CENIDE), University of Duisburg-Essen, 47057 Duisburg, Germany; [orcid.org/0000-0001-8395-3541](https://orcid.org/0000-0001-8395-3541)*

**Jörg Behler** – *Lehrstuhl für Theoretische Chemie II, Ruhr-Universität Bochum, 44780 Bochum, Germany; Research Center Chemical Sciences and Sustainability, Research Alliance Ruhr, 44780 Bochum, Germany; [orcid.org/0000-0002-1220-1542](https://orcid.org/0000-0002-1220-1542)*

Complete contact information is available at: <https://pubs.acs.org/doi/10.1021/acs.jpcc.5c08004>

## Notes

The authors declare no competing financial interest.

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